

Imperfections – deformation and microstructures in polycrystals

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8- Textures in metals

8- Textures in metals a- Typical tests

Textures in polycrystals

- Complex phenomenon
- Need for multi-scale modeling
- Very dependent on material history

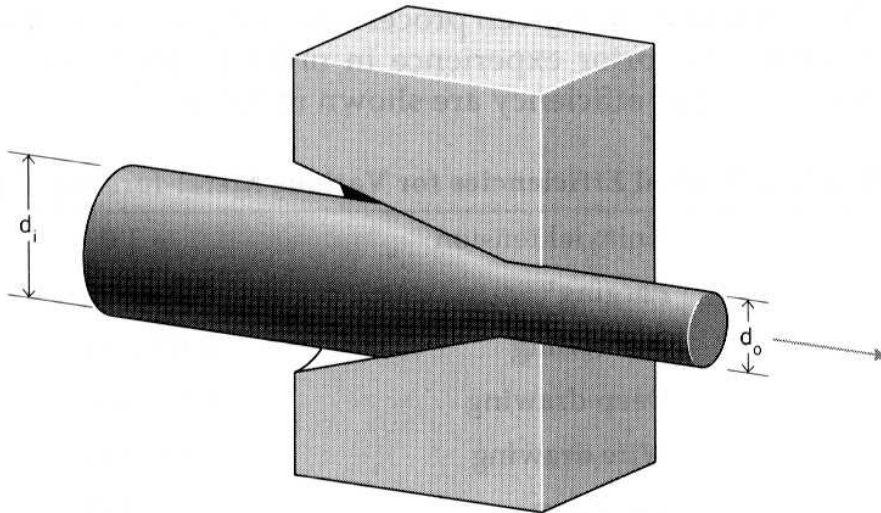
If deformation experiment is too complicated

- Complex textures
- Difficult to understand

Therefore, we often rely on a short set of typical tests

- Compression
- Tension
- Rolling
- And others

Tensile test / wire drawing

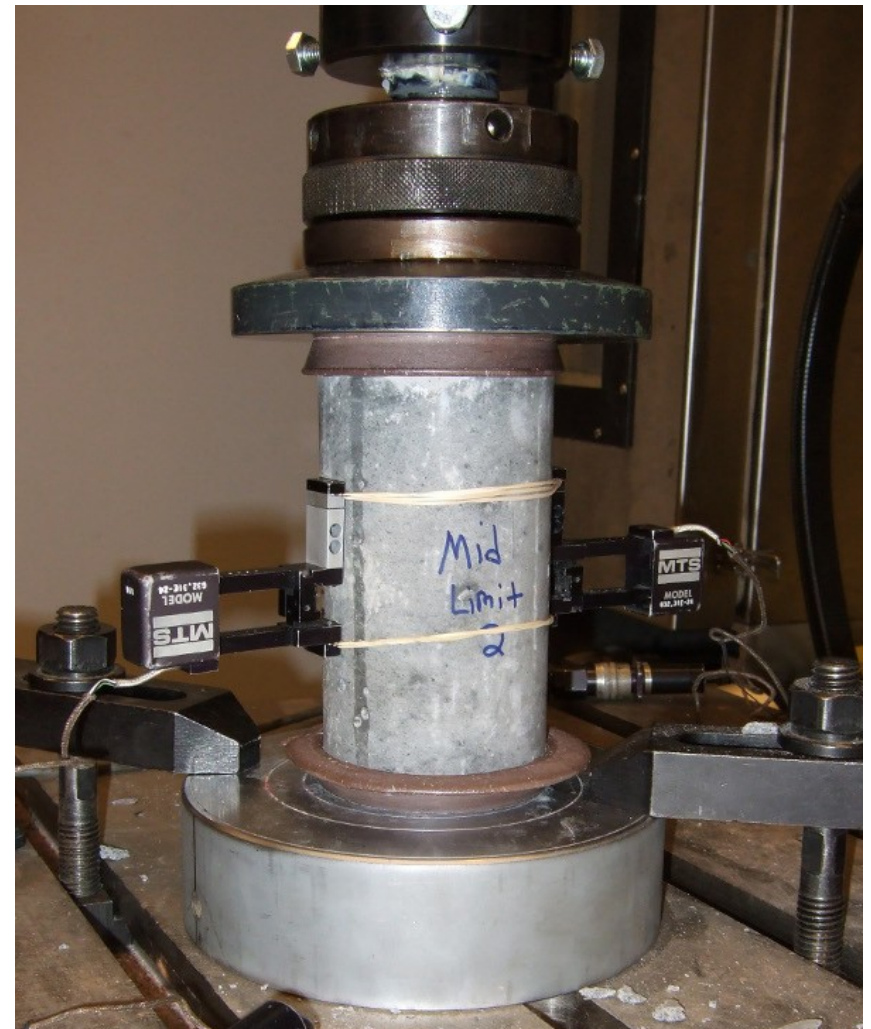


$$\epsilon = \begin{bmatrix} -\Delta/2 & 0 & 0 \\ 0 & -\Delta/2 & 0 \\ 0 & 0 & +\Delta \end{bmatrix}$$

Images :
- Wikipedia
- A. Rollet

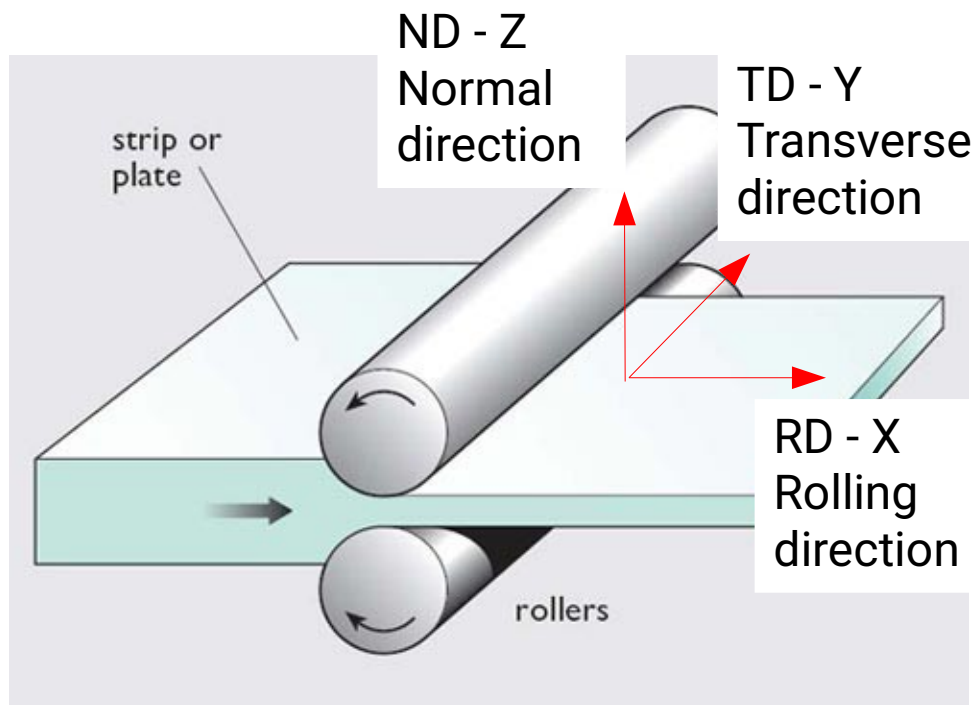


Compression



$$\epsilon = \begin{bmatrix} \Delta/2 & 0 & 0 \\ 0 & \Delta/2 & 0 \\ 0 & 0 & -\Delta \end{bmatrix}$$

Rolling



$$\epsilon = \begin{bmatrix} \Delta & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & -\Delta \end{bmatrix}$$

8- Textures in metals *b*- Face centered cubic

Slip systems - fcc

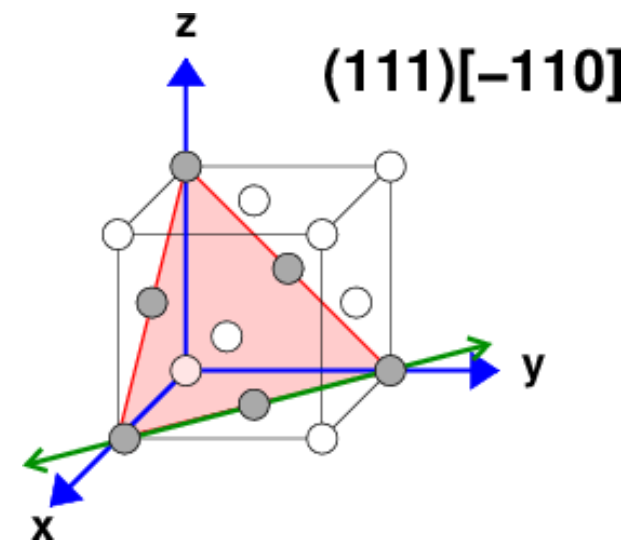
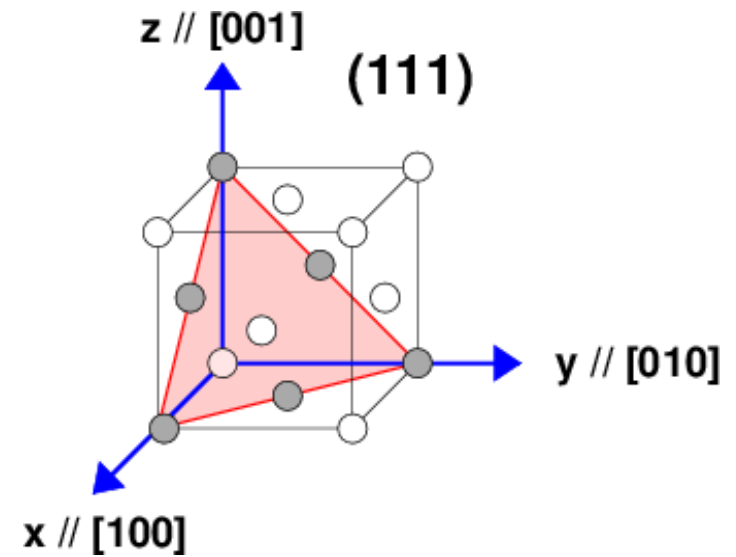
Slip systems in fcc metals:

- slip direction: $\langle 110 \rangle$
- slip plane: $\{111\}$

System: $(111)[-110]$

4 planes $\{111\}$: (111) , $(\bar{1}11)$, $(1\bar{1}1)$, $(11\bar{1})$;

- For each: 3 directions $\langle 110 \rangle$
- 12 systems, that can each operate in 2 directions (+ ou -).



Fcc - Compression Textures

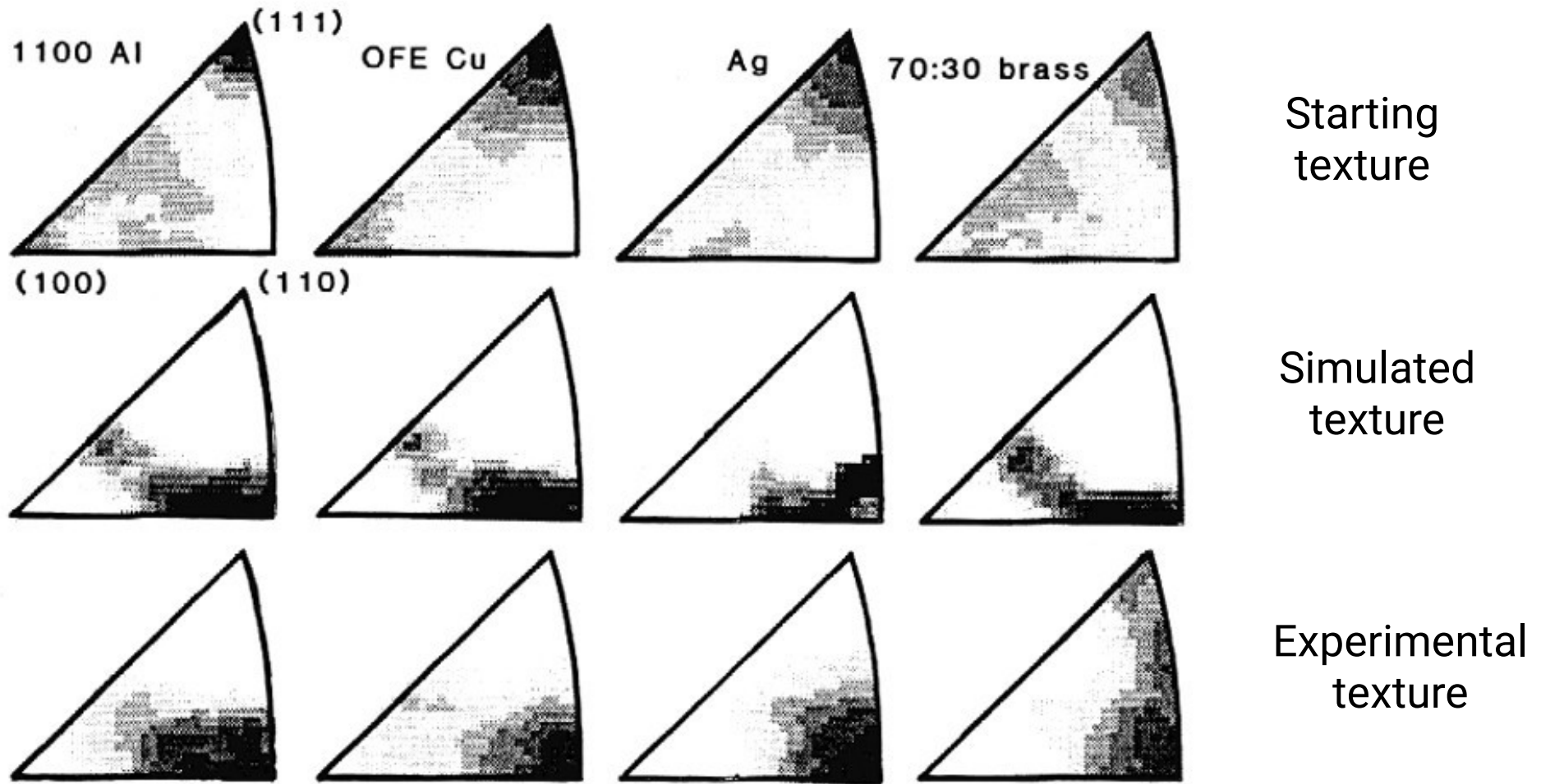
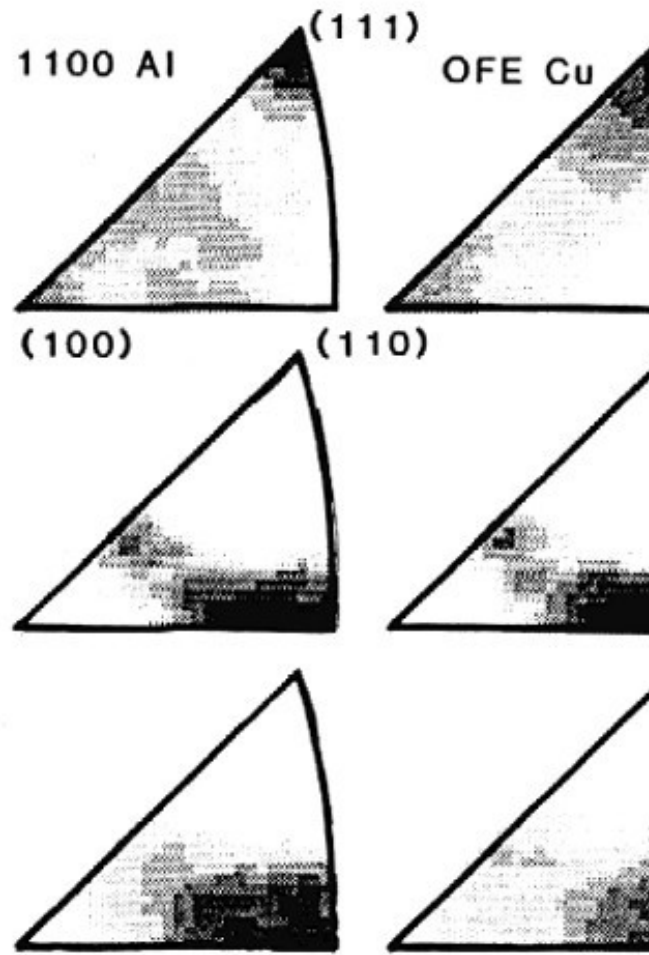


Image Kocks, Tomé, Wenk, 1998, Ch. 5

Fcc - Compression Textures



Quickly:

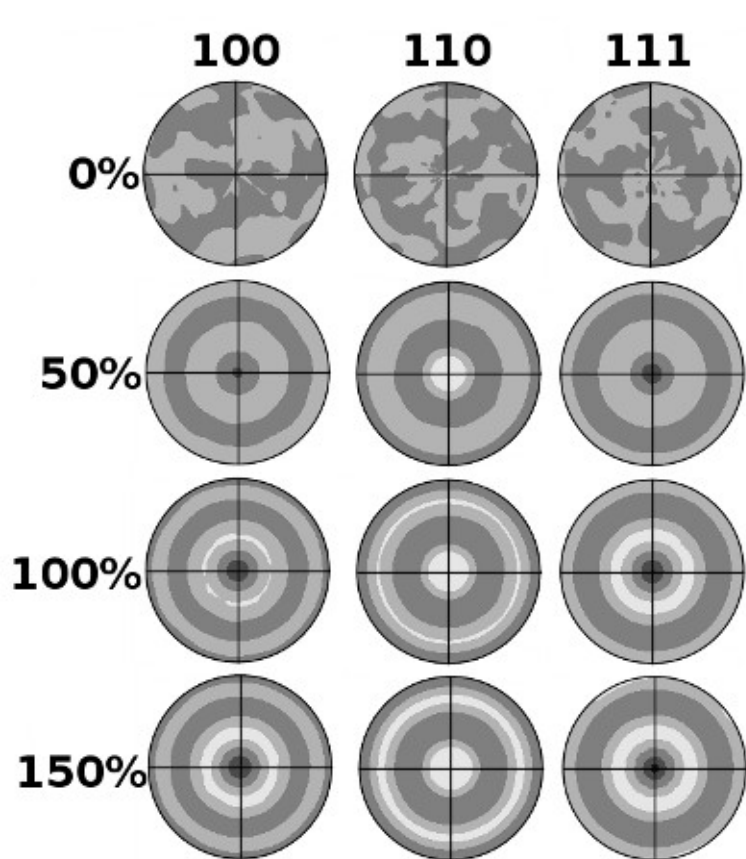
- Fcc metals in compression
- Maximum at 110
- (110) planes with a high probability of begin perpendicular to compression

Experimental texture

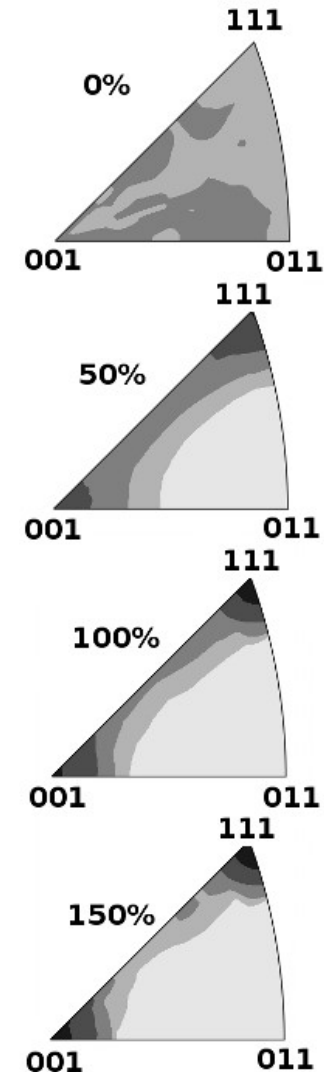
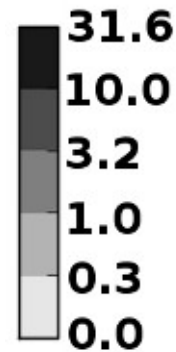
Image Kocks, Tomé, Wenk, 1998, Ch. 5

Fcc - Drawing Textures

Textures simulations for drawing of fcc metal



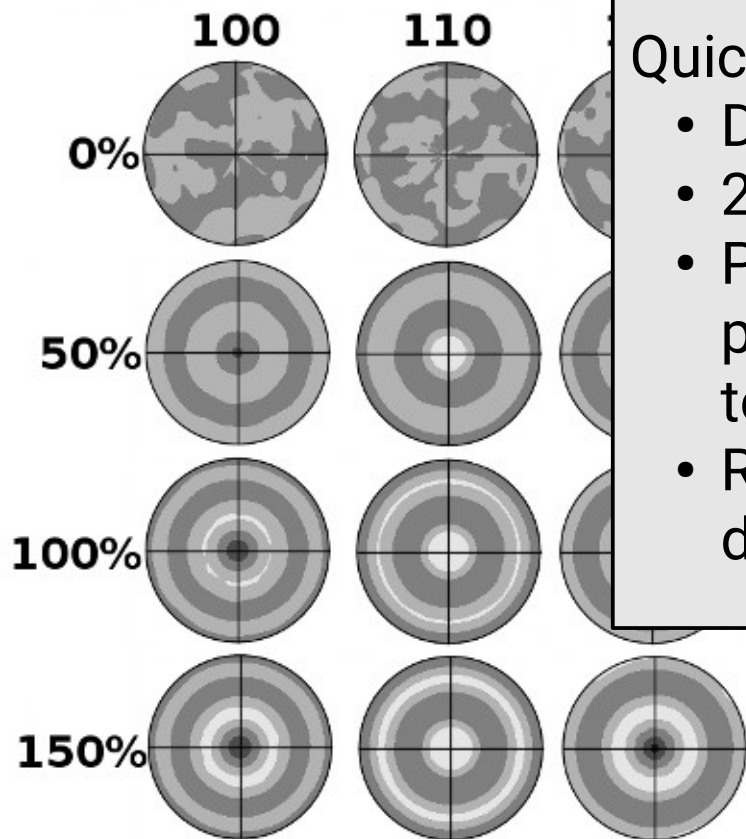
Pole figures (tension direction at the center)



Inverse pole figures of the tension direction

Fcc - Drawing Textures

Textures simulations for drawing of fcc metal

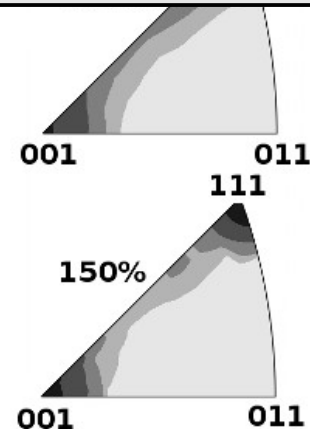


Pole figures (tension direction at the center)

Quickly:

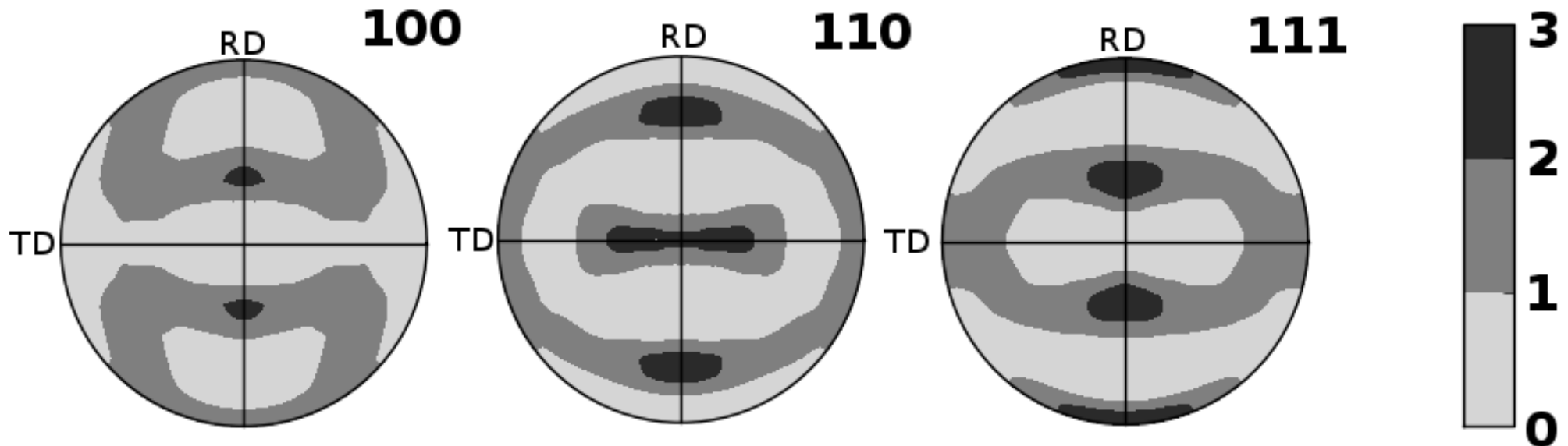
- Drawing of fcc metals
- 2 maxima at 100 and 111
- Planes (100) and (111) with a high probability to be perpendicular to the tension direction
- Relative weight of 100 and 111 depends on the material.

0.3
0.0



Inverse pole figures of the tension direction

Fcc - Rolling texture



Texture simulation after 50 % rolling for a metal with the fcc structure

Rough characteristics :

- (110) planes // rolling plane
- $\langle 1\bar{1}2 \rangle$ // rolling direction

In reality, one often compares the obtained texture to classical cases (copper, brass, etc).

8- Textures in metals c- Centered cubic

Slip systems - bcc

Slip systems in the bcc structure:

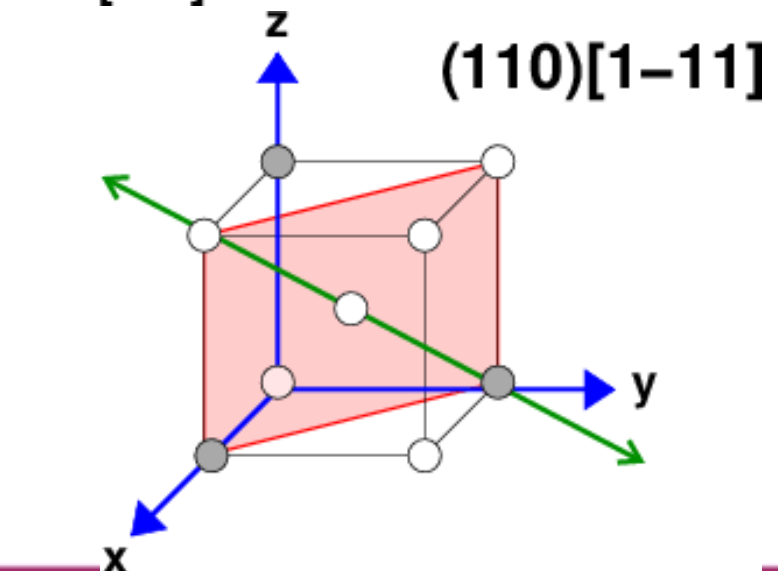
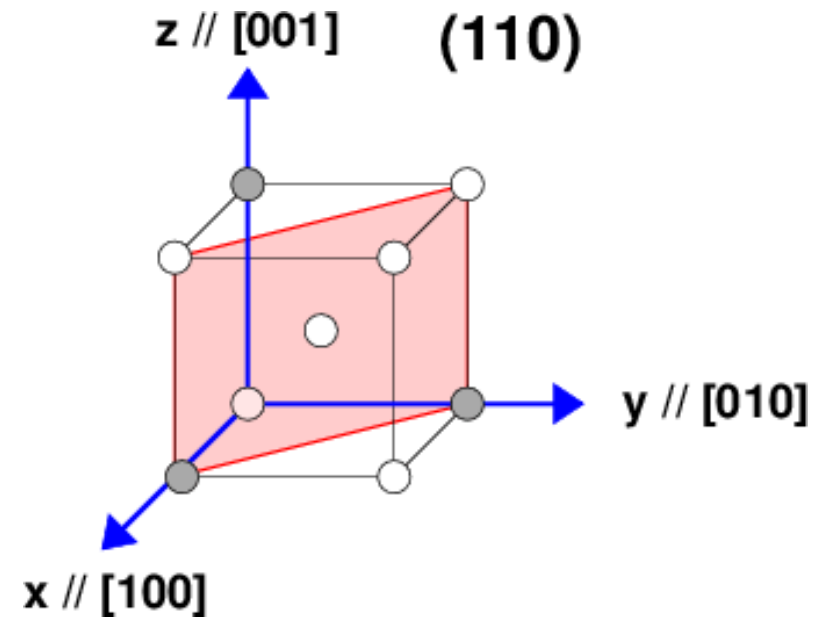
- Slip direction: $\langle 111 \rangle$
- Slip plane: $\{110\}$

Slip system: $(110)[1-11]$

Other possible slip planes but slip direction is most-often $\langle 111 \rangle$.

6 $\{110\}$ planes: (110) , (011) , (101) , $(\bar{1}\bar{1}0)$, $(0\bar{1}\bar{1})$, $(10\bar{1})$;

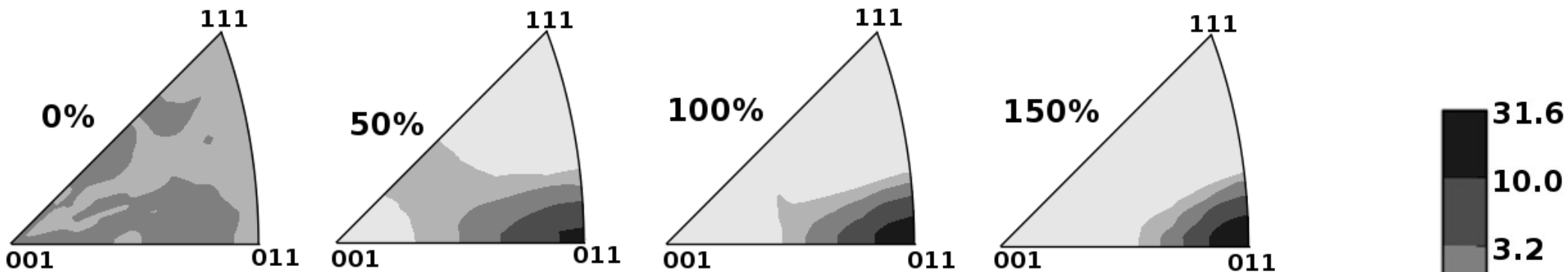
- For each: 2 $\langle 111 \rangle$ directions,
- 12 slip systems, each can operate in 2 directions (+ or -).



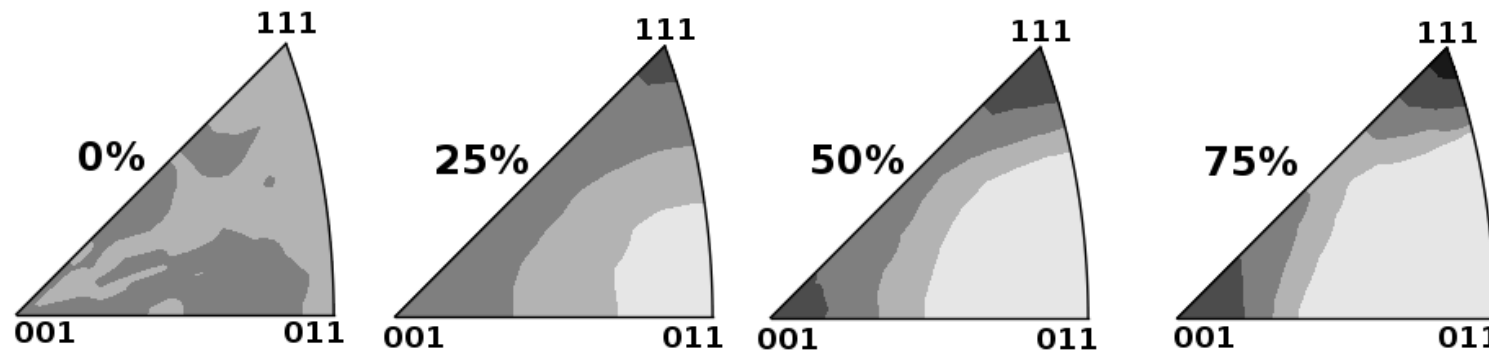
Bcc- Drawing and Compression

Simulation of drawing texture for a bcc metal

Same as fcc metal in compression



Simulation of a compression texture for a bcc metal



Same as drawing texture for fcc

Fcc/ bcc Comparison

Slip systems:

- Fcc : (111)[-110]
- Bcc : (110)[1-11]

Drawing textures:

- Fcc : 2 maxima, in 001 and 111, minimum in 011
- Bcc : maximum in 011

Compression textures:

- Fcc : maximum in 011
- Bcc : 2 maxima, in 001 and 111, minimum in 011

Rolling textures

- More complex.
- You can switch between fcc and bcc metals rolling textures by inverting the RD and ND directions in the projection.

Fcc in compression

~ bcc drawing

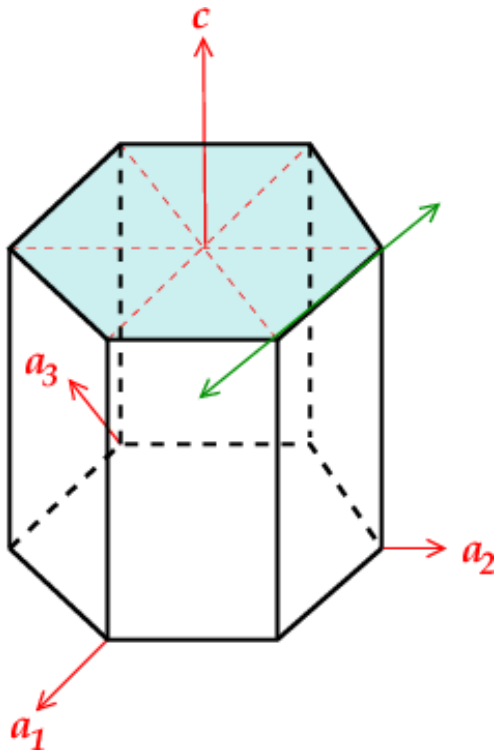
Fcc drawing

~ bcc in compression

8- Textures in metals c- Hexagonal closed packed

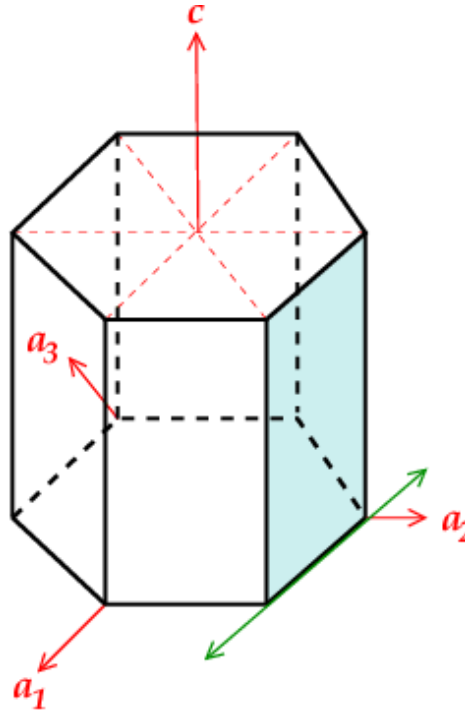
Hcp Slip Systems: $\langle 11\bar{2}0 \rangle$

Slip slip systems with $\langle 11\bar{2}0 \rangle$ Burgers vector
also called $\langle a \rangle$



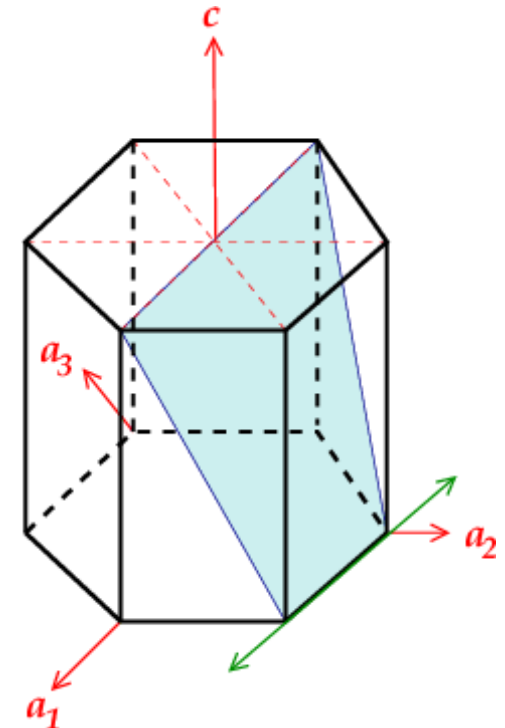
Basal slip
 $(0001)[11\bar{2}0]$

3 equivalent systems



Prismatic slip
 $(01\bar{1}0)[11\bar{2}0]$

3 equivalent systems

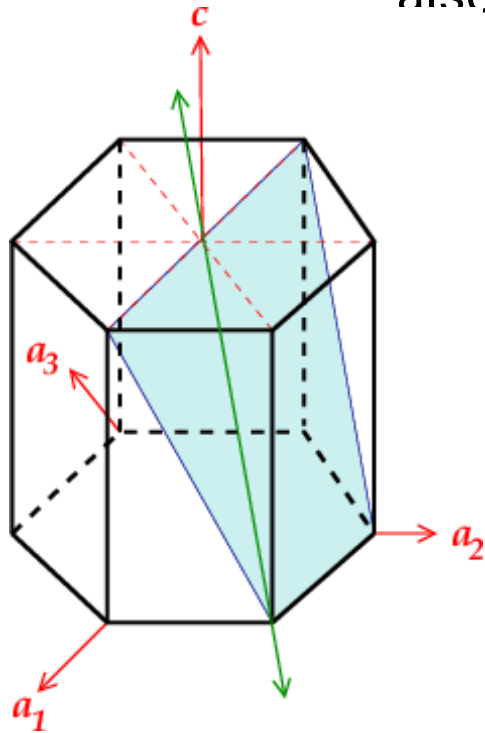


Pyramidal $\langle a \rangle$ slip
 $(01\bar{1}1)[11\bar{2}0]$

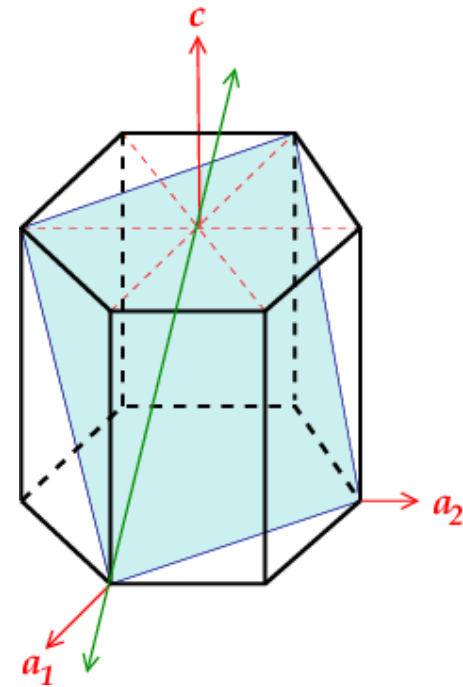
6 equivalent systems

Hcp Slip Systems: $\langle 11\bar{2}3 \rangle$

Slip slip systems with $\langle 11\bar{2}3 \rangle$ Burgers vector
also called $\langle c+a \rangle$



Pyramidal $\langle c+a \rangle$ slip
First order
 $(10\bar{1}1)[11\bar{2}3]$
12 equivalent systems

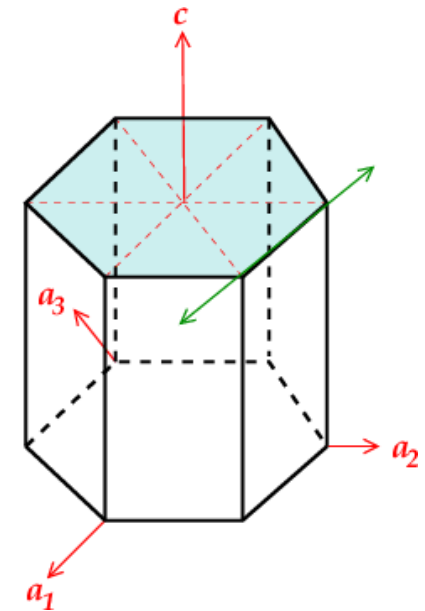
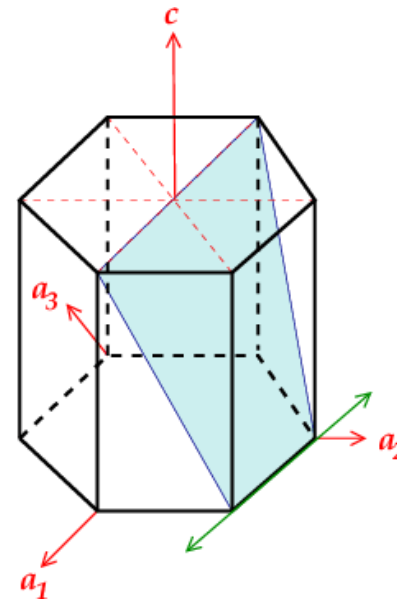
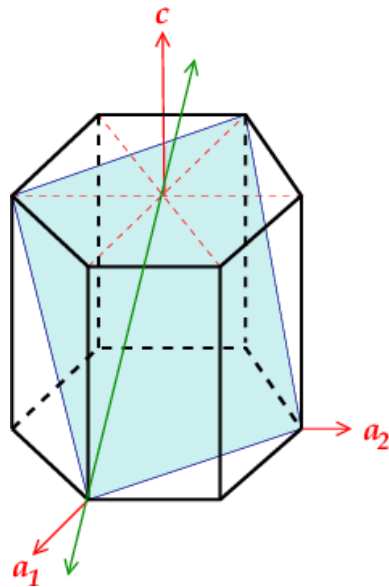


Pyramidal $\langle c+a \rangle$ slip
Second order
 $(11\bar{2}2)[11\bar{2}3]$
6 equivalent systems

$$c/a > 1.633$$

Hexagonal metals with $c/a > 1.633$

- Cd, Zn
- Basal slip
- Pyramidal $\langle a \rangle$ slip
- Pyramidal $\langle c+a \rangle$ second order
- Twins



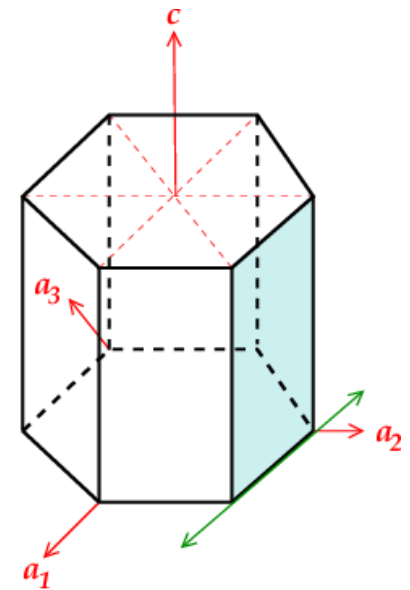
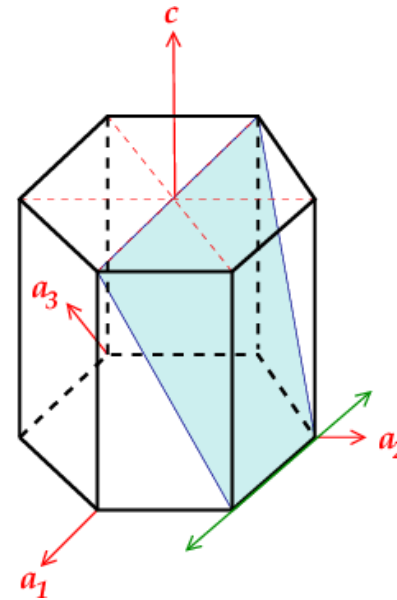
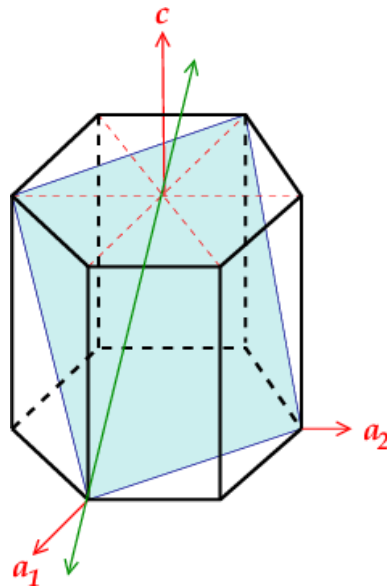
Notes :

- Basal slip is dominant
- Twinning is very important

$$c/a < 1.633$$

Hexagonal metals with $c/a < 1.633$

- Zr, Ti, Hf ;
- Prismatic slip,
- Pyramidal $\langle a \rangle$ slip,
- Pyramidal $\langle c+a \rangle$ second order,
- Twins.



Notes:

- You can also have basal slip
- Twinning is very important

Hcp - Compression / Extrusion Texture

Pure Titanium

Extrusion:

- Maximum at $10\bar{1}0$

Compression:

- Maximum close to 0001
- 25° of 0001 , $\sim \langle 11\bar{2}4 \rangle$

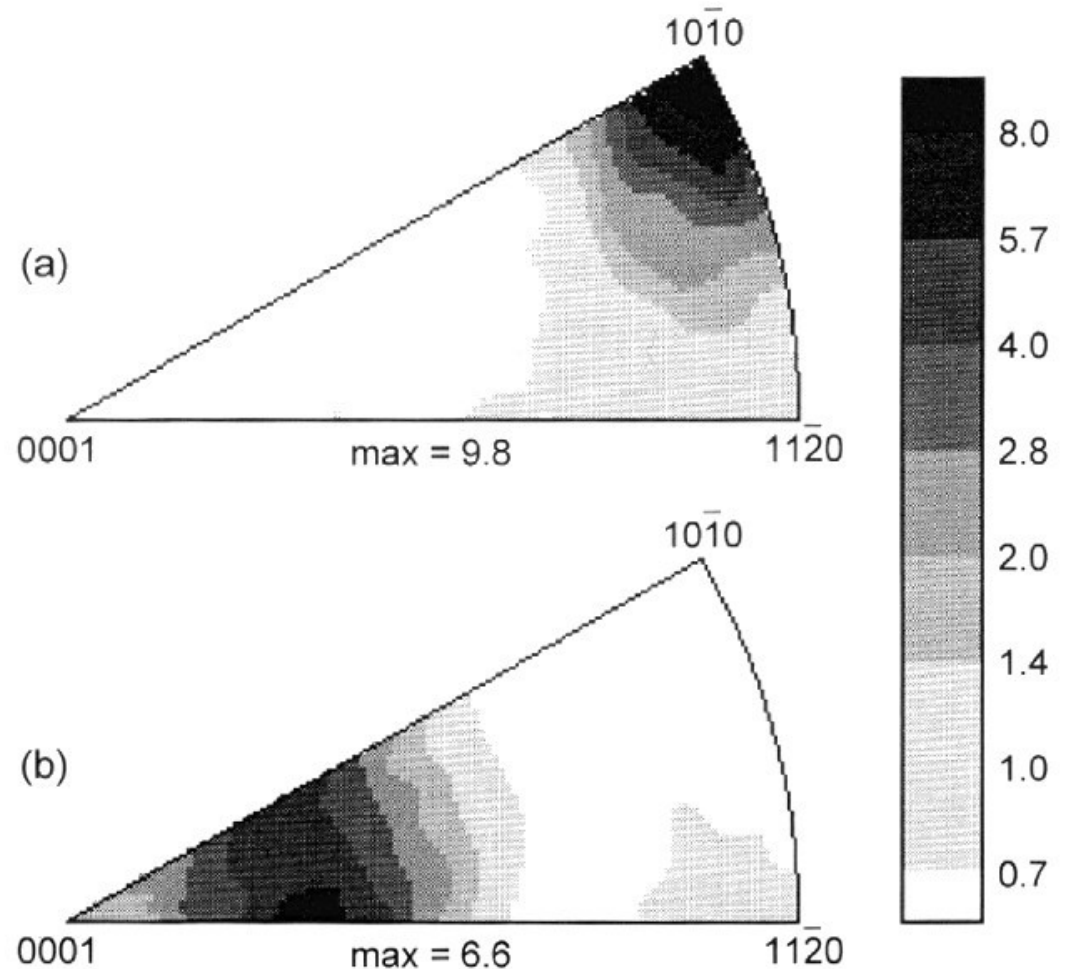


Fig. 20. Inverse pole figures of pure titanium: (a) extruded to a von Mises equivalent strain of 1.75 (extrusion-axis inverse pole figure), (b) forged and cross-rolled to a von Mises equivalent strain of 1.98 (plate normal inverse pole figure).

Comparison Titanium / Zirconium

General rule for hcp metals:

- Complex behavior due to activity of twinning modes
- Simple consideration on slip systems not always relevant

Compression Zr

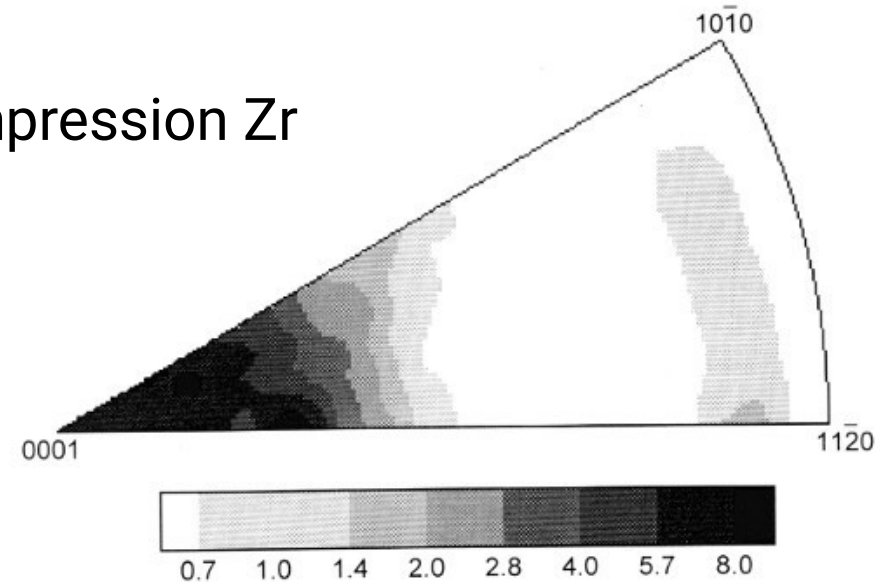
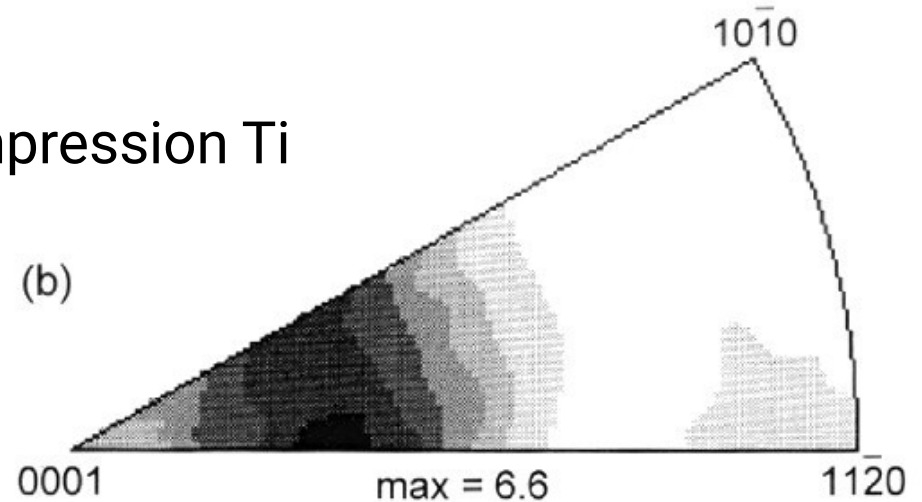


Fig. 21. Inverse pole figure (plate normals) for forged and cross-rolled zirconium.

Compression Ti



Hcp - Rolling Textures

$c/a > 1.633$:

0001 splits in two components in the plane between rolling normal and rolling direction

$c/a = 1.633$:

0001 perfectly aligned with rolling plane normal

$c/a < 1.633$:

0001 split in two components in the plane formed by the transverse and normal directions

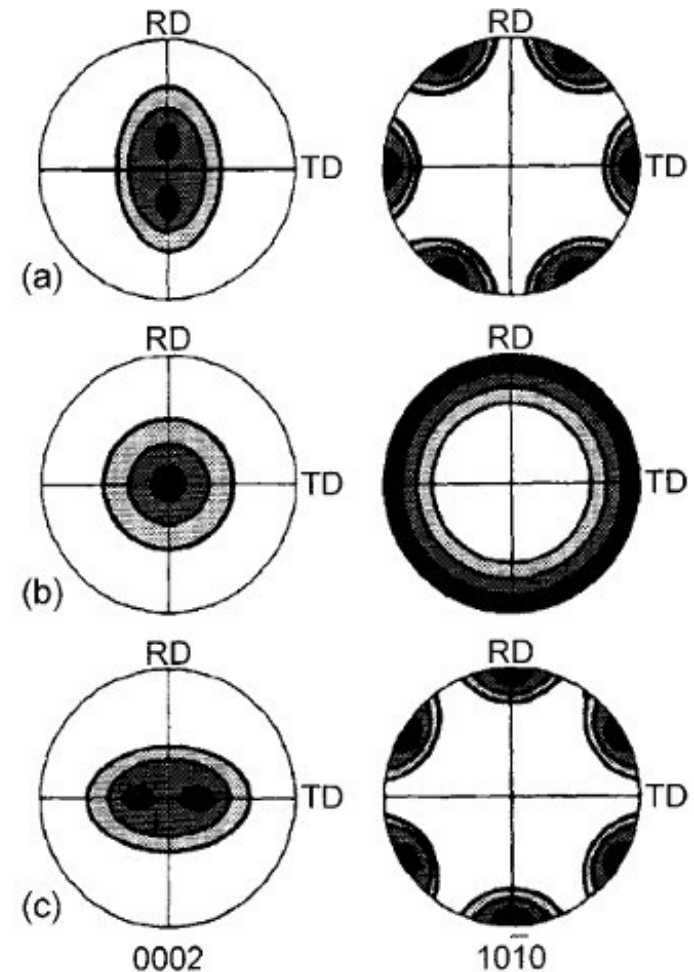


Fig. 22. Schematic rolling textures in hcp metals with c/a ratios of (a) greater than 1.633, (b) approximately equal to 1.633 and (c) less than 1.633. 0002 and $10\bar{1}0$ pole figures. [TENCKHOFF 1988].